

Crack closure and retardation effects – experiments and modelling

ESIS TC24 Reliability of Railway Structures, Leoben, 24.-25.10.2016

J. Maierhofer, H.-P. Gänser





"Safe and economic operation of railway axles" (**EBFW2**) Contents:

- Development of a computational method for determining the residual life and inspection intervals of railway axles by means of fatigue crack growth calculations
- Determination of fracture mechanics material parameters for axle materials
- Validation of the computational model by means of laboratory specimens and component tests on the scales 1:3 and 1:1
- Measurement of load spectra (locomotive and passenger car)
- Differences between computation and 1:1 test results showed the need for additional research
 → EBFW3

Main goal (EBFW3): Clarification of the differences between laboratory specimen and 1:1 tests (or real components, respectively)





Current state of knowledge:

- Crack growth in real components often over-conservative
- Behaviour of short cracks¹



• Growth of cracks emanating from deep sharp notches²



Even more conservative in predicting crack propagation!!

¹ Maierhofer et al., *Modified NASGRO equation for physically short cracks,* Int. J. Fatigue, 2014. ² Maierhofer et al., *Modified Kitagawa-Takahashi diagram accounting for finite notch depths,* Int. J. Fatigue, 2014.



Mechanisms which can lead to a drop in the crack propagation rate¹:

- a) plasticity induced crack closure,
- b) oxide induced crack closure,
- c) roughness induced crack closure,
- d) fluid induced crack closure,
- e) transformation induced crack closure,
- f) crack branching,
- g) crack bridging due to fibers or
- h) due to particles [Suresh, Ritchie]



¹ Suresh, *Fatigue of Materials*, Cambridge University Press 1998.



Investigated material: QT steel, $R_m \sim 700$ MPa



- Notch generation by wire electrical discharge machining
- Notch sharpening by razor blade cutting









Eroded notch with razor blade cut

Razor blade cutting rig



Introduction / Experimental setup

Resonant testing rig (RUMUL TESTRONIC)



8-point bending mounting



- Compression precracking $(R = 20, \Delta K \approx 14 \text{ MPa}\sqrt{\text{m}})$
- No crack closure effects at the start of an experiment
- Crack propagation measurement: direct current potential drop technique



Four main reasons for difference between laboratory specimen and 1:1 tests

1. Residual stresses



3. "small" loads



2. Overloads



4. Downtime





Crack retardation due to residual stresses



Difference laboratory specimen / 1:1 test:

Laboratory specimen mostly free from residual stresses!

Required for crack retardation: Residual compressive stresses

Origin of residual stresses in 1:1 tests and in real components:



- metal forming
- heat treatment
- machining

• • •

Surface treatment

- shot peening
- deep rolling
- laser shock treatment

•

• • •



Crack retardation due to residual stresses / experiments

Influence of residual compressive stresses



¹ Maierhofer et al., *Modified NASGRO equation for short cracks and application to the fitness-for-purpose assessment of surface-treated components,* Proceedings ECF20, 2014. 9



Influence of residual compressive stresses (example: SEB specimen¹):



¹ Maierhofer et al., Modified NASGRO equation for short cracks and application to the fitness-for-purpose assessment of surfacetreated components, Proceedings ECF20, 2014.



Consider residual stress fields in computational models by changing the mean stress:



$$\Delta K = K_{\max}(\sigma_{\max}) - K_{\min}(\sigma_{\min})$$

...and calculate the corresponding crack growth rate

 $\frac{da}{dN} \propto \Delta K^m$



Four main reasons for difference between laboratory specimen and 1:1 tests

1. Residual stresses



3. "small" loads



2. Overloads



4. Downtime





Crack retardation due to overloads



Difference laboratory specimen / real component:

No constant load amplitude!



Required for lifetime enhancement : Tension overload



Influence of a single tensile / compressive overload

reference specimen:

- constant load (ΔF =const.), R=-0,5
- start with $\Delta K = 14 \text{ MPaVm}$
- no overload





Influence of a single compressive overload

Compressive overload:

- constant load (Δ F =const.), R=-0,5
- start with $\Delta K = 14$ MPaVm until $\Delta K = 18$ MPaVm
- compressive overload 3 times higher than the primary load amplitude





Influence of a single compressive overload

Compressive overload:

- constant load (∆F =const.), R=-0,5
- start with $\Delta K = 14$ MPaVm until $\Delta K = 18$ MPaVm
- compressive overload 3 times higher than the primary load amplitude
- again apply primary load amplitude until final fracture of the specimen





Influence of a single tensile overload

Tensile overload:

- constant load (ΔF =const.), R=-0,5
- start with $\Delta K = 14$ MPaVm until $\Delta K = 18$ MPaVm
- tensile overload 3 times higher than the primary load amplitude





Influence of a single tensile overload

Tensile overload :

- constant load (Δ F =const.), R=-0,5
- start with $\Delta K = 14$ MPaVm until $\Delta K = 18$ MPaVm
- tensile overload 3 times higher than the primary load amplitude
- again apply primary load amplitude until final fracture of the specimen





Influence of overloads on the crack growth behaviour of long cracks:

Compressive overload: Tensile overload:

no influence on crack growth / lifetime significant retardation of the crack propagation rate; increase of lifetime





Approximate description of overload induced crack retardation by means of the Willenborg-Gallagher-Hughes model:

• effective load ratio:

$$R_{\rm eff} = \frac{K_{\rm min} - K_{\rm r}}{K_{\rm max} - K_{\rm r}}$$

• SIF due to overload induced residual stresses in front of the crack tip:

$$K_{\rm r} = \mathbf{\Phi} \cdot \left[K_{\rm max,OL} \left(1 - \frac{\Delta a}{Z_{\rm OL}} \right)^{1/2} - K_{\rm max} \right]$$

• Size of overload influence zone depends on overload:

$$Z_{\rm OL} = L_{\rm OL} (K_{\rm max} - \Delta K_0)^{p_{\rm OL}}$$



Four main reasons for difference between laboratory specimen and 1:1 tests

1. Residual stresses



3. "small" loads



2. Overloads



4. Downtime





Crack retardation due to "small" loads





Required for lifetime enhancement : **quite a few small loads**



Influence of small loads

reference specimen:

- constant load (ΔF =const.), R=-0,5
- start with $\Delta K = 14 \text{ MPaVm}$
- no small loads → no build-up of oxide layer





Influence of small loads

Build-up of oxide layer:

- constant load (ΔF =const.), R=-0,5
- start with $\Delta K = 14$ MPaVm until $\Delta K = 18$ MPaVm
- build-up of oxide layer with 9 MPaVm (\approx 34·10⁶ cycles)





Influence of small loads

Build-up of oxide layer:

- constant load (∆F =const.), R=-0,5
- start with $\Delta K = 14$ MPaVm until $\Delta K = 18$ MPaVm
- build-up of oxide layer with 9 MPaVm ($\approx 34.10^6$ cycles)
- again apply primary load amplitude until final fracture of the specimen Build-up of





Investigation of two specimens using an IonSlicer specimen a: $\Delta K_{oxide} = 7$ MPaVm, 200.000 load cycles specimen b: $\Delta K_{oxide} = 9$ MPaVm, 33.000.000 load cycles

Specimen preparation:





Crack retardation due to small loads / fracture surface analysis

Working principle of an IonSlicer:



EHT = 20.00 kV

WD = 8.5 mm

Signal A = CZ BSD

Mag = 3.00 K X

Date :22 Sep 2015

me = Probe 224 Pos2



Comparison of the two oxide layers:

- **X** oxide layer of specimen b significantly thicker
- \rightarrow thickness of oxide layer depending on
 - 1. number of applied load cycles and
 - 2. load amplitude (of the small loads)







The area of influence Z_{0x} of small loads and therefore the thickness of the oxide layer depends on the number of load cylces and the corresponding load amplitude :

$$Z_{\rm ox} = L_{\rm ox} \left(1 + R\right)^{s_{\rm ox}} \Delta K_{\rm ox}^{p_{\rm ox}} N_{\rm ox}^{q_{\rm ox}}$$

Modelling of the oxide induced crack retardation by increasing the crack growth threshold:

$$\Delta K_{\rm th,ox} = K_{\rm ox} (1+R)^{r_{\rm ox}} \Delta K_{\rm ox}^{m_{\rm ox}} N_{\rm ox}^{m_{\rm ox}}$$



Four main reasons for difference between laboratory specimen and 1:1 tests

1. Residual stresses



3. "small" loads



2. Overloads



4. Downtime





Crack retardation due to downtime











- constant load (ΔF =const.), R=-1
- start with $\Delta K = 14 \text{ MPaVm}$
- after $\Delta K = 17,2$ MPaVm was reached, experiment was stopped, the specimen removed, again clamped in the testing rig and restarted with primary load amplitude





- constant load (ΔF =const.), R=-1
- start with $\Delta K = 14 \text{ MPaVm}$
- after $\Delta K = 17,2$ MPaVm was reached, experiment was stopped and the specimen removed
- after two weeks in ambient air the experiment was again clamped in the testing rig and restarted with the primary load amplitude





- constant load (ΔF =const.), R=-1
- start with $\Delta K = 14 \text{ MPaVm}$
- after $\Delta K = 17,2$ MPaVm was reached, experiment was stopped and the specimen removed
- after two weeks in water the experiment was again clamped in the testing rig and restarted with the primary load amplitude





- constant load (ΔF =const.), R=-1
- start with $\Delta K = 14 \text{ MPaVm}$
- after $\Delta K = 17,2$ MPaVm was reached, experiment was stopped and the specimen removed
- after two weeks in liquid nitrogen the experiment was again clamped in the testing rig and restarted with the primary load amplitude





Crack retardation due to downtime / fracture surface analysis



36



Four main reasons for difference between laboratory specimen and 1:1 tests (or real components, respectively):

- 1. Residual stresses
 - Residual compressive stresses decrease the crack growth rate and lead to higher lifetimes
 - In small scale specimens there are often no residual stresses left

2. Overloads

- Tensile overloads can lead to a significant retardation of crack growth
- Compressive overloads have no effects (for long cracks and small scale yielding)

3. Small loads

- Depending on the applied load and number of cyles a build-up of oxide layer can occur
 → crack closure increases and therefore the crack growth rate decreases
- Small loads are often not considered during testing ("omission")

4. Downtime

• Influence on crack growth in the near-threshold region



Financial support by the Austrian Federal Government (in particular from Bundesministerium für Verkehr, Innovation und Technologie and Bundesministerium für Wissenschaft, Forschung und Wirtschaft) represented by Österreichische Forschungsförderungsgesellschaft mbH and the Styrian and the Tyrolean Provincial Government, represented by Steirische Wirtschaftsförderungsgesellschaft mbH and Standortagentur Tirol, within the framework of the COMET Funding Programme is gratefully acknowledged.











Dr. Jürgen Maierhofer

Materials Center Leoben Forschung GmbH (MCL) Roseggerstraße 12 A-8700 Leoben, Austria Tel.: +43 3842 45922 - 41 Email: juergen.maierhofer@mcl.at www.mcl.at